

Short Communication

# Novel Low-Permittivity, Low-Sintering-Temperature $\text{Na}_2\text{WO}_4$ Microwave Dielectric Ceramics for LTCC Applications

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## Abstract

A novel  $\text{Na}_2\text{WO}_4$  ceramic with low sintering temperature was successfully synthesized by means of the reactive sintering process. The phase composition, microstructure and microwave dielectric properties of the  $\text{Na}_2\text{WO}_4$  dielectric ceramic with different sintering temperature (600, 650, 680 and 700 °C) were investigated in detail using X-ray diffraction (XRD), scanning electron microscopy (SEM) and a vector network analyzer, respectively. Excellent microwave dielectric properties with a relative permittivity ( $\epsilon_r$ ) ~ 3.45, a quality factor ( $Q \times f$ ) ~ 38 244 GHz (at 13.64 GHz) and a temperature coefficient of resonant frequency ( $\tau_f$ ) ~ -42 ppm/K was obtained at a sintering temperature of 680 °C for 2 h. The results indicate that the  $\text{Na}_2\text{WO}_4$  ceramic is a promising candidate for low-temperature co-fired ceramic technology.

*Keywords:* Microwave dielectric properties, low sintering temperature,  $\text{Na}_2\text{WO}_4$ , reactive sintering process

## I. Introduction

Microwave dielectric materials with light weight, low cost, high dielectric constant and low loss have been widely applied in the ever-growing wireless communication and microwave circuits, such as dielectric resonators, filters, duplexers, antennas, oscillators and smart phones<sup>1–3</sup>. Low-temperature-sintered ceramics with a high quality factor ( $Q \times f$ ), an appropriate dielectric constant ( $\epsilon_r$ ), and a near-zero temperature coefficient of resonant frequency ( $\tau_f$ ) have attracted much interest in low-temperature co-fired ceramics (LTCC) applications<sup>4–8</sup>. Meanwhile a large number of low-temperature-sintered materials with excellent microwave dielectric properties have been reported, such as glasses and glass-ceramics, vanadate ceramics, tungstate ceramics, molybdate ceramics, tellurate ceramics and borate ceramics<sup>8</sup>. Recently, tungstate ceramics have been favored by researchers because of their low dielectric constant, low loss, and ultra-low sintering temperature. Bowen Zhang et al. obtained excellent microwave dielectric properties with a  $\epsilon_r$  of 10.69, a  $Q \times f$  value of 20 000 GHz (at 10 GHz) and a  $\tau_f$  value of -35 ppm/K for  $0.97\text{CoWO}_4-0.03\text{CuO}$  sintered at 950 °C with the use of CuO to modify the  $\text{CoWO}_4$  ceramics<sup>1</sup>. Di Zhou et al. reported that the  $\text{Li}_2\text{WO}_4$  ceramic with phenacite structure can be sintered effectively at 640–660 °C with the solid-state reaction method, and obtained a  $\epsilon_r$  of 5.5, a  $Q \times f$

value of 62 000 GHz and a  $\tau_f$  value of -146 ppm/K at 15.7 GHz<sup>9</sup>. Jie Li et al. reported a  $\text{Li}_4\text{WO}_5$  low-firing ceramic with rock salt structure and excellent microwave dielectric properties with a near-zero temperature coefficient of resonant frequency ~ -2.6 ppm/K, a relative permittivity ~ 8.6 and a quality factor ~ 23 100 GHz (at 11.0 GHz)<sup>10</sup>. Huanfu Zhou et al. studied the microwave dielectric properties of  $\text{LiBiW}_2\text{O}_8$  ceramics with low sintering temperature, the results indicate that  $\text{LiBiW}_2\text{O}_8$  ceramic with a  $\epsilon_r$  of 26.5, a  $Q \times f$  value of 16 400 GHz, and a  $\tau_f$  value of 70 ppm/K could be prepared at 700 °C for 2 h<sup>11</sup>. Liang Fang et al. obtained the low-firing  $\text{Li}_2\text{Mg}_2\text{W}_3\text{O}_{12}$  ( $\epsilon_r \sim 8.4$ ,  $Q \times f \sim 56 700$  GHz,  $\tau_f \sim -72.8$  ppm/K) and  $\text{Li}_2\text{Zn}_2\text{W}_3\text{O}_{12}$  ( $\epsilon_r \sim 11.3$ ,  $Q \times f \sim 24 500$  GHz,  $\tau_f \sim -100$  ppm/K) ceramics at 720 and 700 °C, respectively<sup>12</sup>. Other tungstate ceramics are summarized in Table 1.

Despite the discoveries of the above-mentioned new classes of microwave dielectric ceramics with ultra-low sintering temperature, the development of such or similar materials with the required high performance is still a highly challenging task, particularly the unreported  $\text{Na}_2\text{WO}_4$  ceramic at present. In this work, a novel microwave substrate  $\text{Na}_2\text{WO}_4$  ceramic was synthesized with the reactive sintering process. Furthermore, the effects of sintering temperature on phase composition, microstructure and microwave dielectric properties were systematically investigated.

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**Table 1:** Comparison of microwave dielectric properties of some tungstate ceramics with microwave dielectric permittivity from No. 5 to No. 15.

No.	Samples	S <sub>T</sub> (°C)	F (GHz)	ε <sub>r</sub>	Q×f (GHz)	τ <sub>f</sub> (ppm/K)	Ref.
1		600	13.7	3.4	13 637	-36.5	
2	Na <sub>2</sub> WO <sub>4</sub>	650	13.7	3.3	16 190	-42.1	This
3		680	13.6	3.5	38 244	-42.1	work
4		700	13.7	3.7	27 049	-48.9	
5	0.97CoWO <sub>4</sub> -0.03CuO	950	10	10.7	20 000	-35	[1]
6	Li <sub>2</sub> WO <sub>4</sub>	650	15.7	5.5	62 000	-146	[9]
7	Li <sub>4</sub> WO <sub>5</sub>	890	11	8.6	23 100	-2.6	[10]
8	LiBiW <sub>2</sub> O <sub>8</sub>	700	6–7	26.5	16 400	-70	[11]
9	Li <sub>2</sub> Zn <sub>2</sub> WO <sub>4</sub>	700	-	11.3	24 500	-100	[12]
10	Li <sub>2</sub> Mg <sub>2</sub> WO <sub>4</sub>	720	-	8.4	56 700	-73	[12]
11	Li <sub>2</sub> Mg <sub>2</sub> W <sub>2</sub> O <sub>9</sub>	820	11	10.3	20 537	-76.9	[17]
12	CaWO <sub>4</sub> -2Li <sub>2</sub> WO <sub>4</sub>	740	10–13	6.1	62 400	-100.1	[18]
13	PbWO <sub>4</sub>	620	7.0	21.4	43 000	-7	[19]
14	Pb <sub>2</sub> WO <sub>4</sub>	520	7.6	16.4	14 800	-95	[19]
15	(Ag, Bi) <sub>0.5</sub> WO <sub>4</sub>	580	7.5	35.9	13 000	-69	[20]

S<sub>T</sub> sintered temperature; F resonant frequency.

## II. Experimental

The Na<sub>2</sub>WO<sub>4</sub> ceramic materials were prepared with the reactive sintering method using proportionate amounts of raw materials: Na<sub>2</sub>CO<sub>3</sub> (≥99.8 %, Sinopharm Chemical Reagent Co., Ltd, Beijing, China) and WO<sub>3</sub> (≥99.0 %, Sinopharm Chemical Reagent Co., Ltd, Beijing, China). The starting materials were accurately weighed based on molar ratios and ball-milled in alcohol solution with ZrO<sub>2</sub> balls for 8 h, and then dried for 24 h. The as-dried powders were mixed with 5 wt% polyvinyl alcohol (PVA) for granulation. The granulated powders were pressed into cylindrical compacts with a diameter of 10 mm and thickness of 5 mm under uniaxial pressure of 220 MPa. Finally, the manufactured green bodies were placed into an alumina crucible and sintered in a temperature range from 600 to 700 °C for 2 h in air atmosphere with a heating rate of 5 K/min, and then cooled down naturally to room temperature inside the furnace.

The structure of the crystalline phase for the as-fabricated Na<sub>2</sub>WO<sub>4</sub> ceramic was identified by means of X-ray diffraction (XRD: D/max-2550V/PC, Rigaku, Japan) with CuKα radiation (λ = 1.5406 Å) at a scanning speed of 10°·min<sup>-1</sup> with 40 kV and 100 mA in the 2θ range of 10–80°. The microstructure of the as-prepared specimens was observed by means of scanning electron microscopy (SEM: FEI-quanta 200, USA). The bulk density of the as-produced samples was measured with the Archimedes method using distilled water as the medium on an electronic balance (Model: XT220A, Precisa, Switzerland) with

0.0001 g precision. A vector network analyzer (ZVB20, Rohde & Schwarz, Germany) was used to test the microwave dielectric properties of the as-fabricated ceramics with the TE<sub>018</sub> shielded cavity method in the frequency range of 9–14 GHz. The temperature coefficients of τ<sub>f</sub> were tested at temperatures ranging from 25 to 80 °C. τ<sub>f</sub> was calculated by the following equation:

$$\tau_f = \frac{f_{80} - f_{25}}{f_{25} \cdot (80 - 25)} \cdot 10^6 \text{ (ppm/K)}$$

where f<sub>25</sub> and f<sub>80</sub> are the TE<sub>018</sub> resonant frequency at 25 and 80 °C, respectively.

## III. Results and Discussion

Fig. 1 shows the room-temperature XRD patterns of the Na<sub>2</sub>WO<sub>4</sub> ceramic sintered at 600–700 °C for 2 h. The

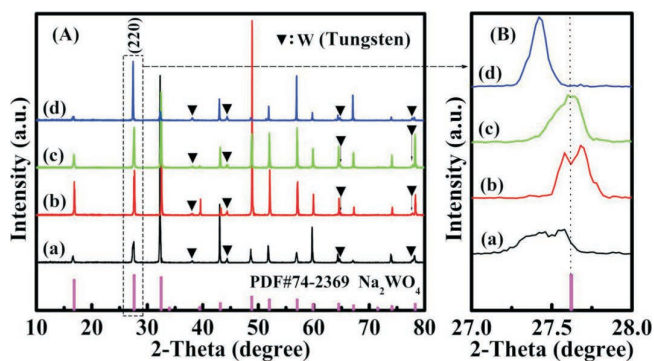


Fig. 1: XRD patterns of Na<sub>2</sub>WO<sub>4</sub> ceramic sintered at different temperature for 2 h, (a) 600 °C, (b) 650 °C, (c) 680 °C, (d) 700 °C.

Na<sub>2</sub>WO<sub>4</sub> phase with a cubic structure belonging to the Fd-3m (227) space group was obtained, but the XRD display also contains some peaks, which was indexed to tungsten (W). The enlarged XRD patterns of the strongest peak ((220) crystal face) in Fig. 1A at the range of 2θ from 27 to 28° are depicted in Fig. 1B. With the increase in sintering temperature, the location of the XRD patterns of the Na<sub>2</sub>WO<sub>4</sub> ceramic shifted to a lower angle. This indicates that the interplanar spacing (D value) of as-prepared Na<sub>2</sub>WO<sub>4</sub> ceramic increased gradually with the rise in sintering temperature.

Fig. 2 presents the SEM images on the surface of Na<sub>2</sub>WO<sub>4</sub> ceramic sintered at different temperatures for 2 h. It could be observed that the grains of the Na<sub>2</sub>WO<sub>4</sub> ceramic gradually formed and grew with the increase in sintering temperature from 600 to 700 °C, and more and more grains could be also observed, especially in the sample sintered at 700 °C. It is clear that uniform and dense grain morphology could be observed in the SEM image of the ceramics sintered at 650 °C (Fig. 2b). The ceramic sintered at 680 °C for 2 h had a dense structure with a fine wavy line size (Fig. 2c). The increase in pores with the sintering temperature (as seen in Fig. 2d) induces the bulk density and the microwave dielectric properties of the ceramics. In order to confirm the compositional difference between two types of grains, composition analysis was performed by means of EDS on the particulate (Spot

B) and non-particulate (Spot A) of the Na<sub>2</sub>WO<sub>4</sub> samples sintered at 680 °C for 2 h (Fig. 2c), and the results are shown in the inset. The ratio of W:Na:O of Spot A is 37.50:36.45:26.05, which was not close to 1:2:4, indicating Spot A was a mix of W phase and Na<sub>2</sub>WO<sub>4</sub> phase with the chemical composition of Na<sub>2</sub>CO<sub>3</sub>-WO<sub>3</sub>. Conversely, the ratio of W:Na:O of Spot B is 20.87:40.44:38.69, which is approximately consistent with composition of Na<sub>2</sub>WO<sub>4</sub>. The EDS analysis of both Spot A and Spot B agreed with the XRD patterns in Fig. 1.

Fig. 3 illustrates the variation of bulk density of Na<sub>2</sub>WO<sub>4</sub> ceramics prepared at different sintering temperature. With the increase in sintering temperature, the bulk density of the specimens increased at first and then decreased slightly. The Na<sub>2</sub>WO<sub>4</sub> sample sintered at an optimum temperature of 650 °C for 2 h exhibits a maximum experimental density of 4.49 g/cm<sup>3</sup>, which corresponds to 88 % of the theoretical density (5.13 g/cm<sup>3</sup>). Therefore, the optimum densification sintering temperature of the Na<sub>2</sub>WO<sub>4</sub> ceramics was considered to be 650 °C for 2 h.

Fig. 4 shows the variations in the values of ε<sub>r</sub>, Q×f and τ<sub>f</sub> of as-prepared Na<sub>2</sub>WO<sub>4</sub> ceramics as a function of sintering temperature. The dielectric constant is dependent on the density, dielectric polarizabilities, and structural character such as the distortion, tilting, and rattling spaces of the oxygen octahedron in the unit cell<sup>13,14</sup>. Generally, the dielectric permittivity should increase with

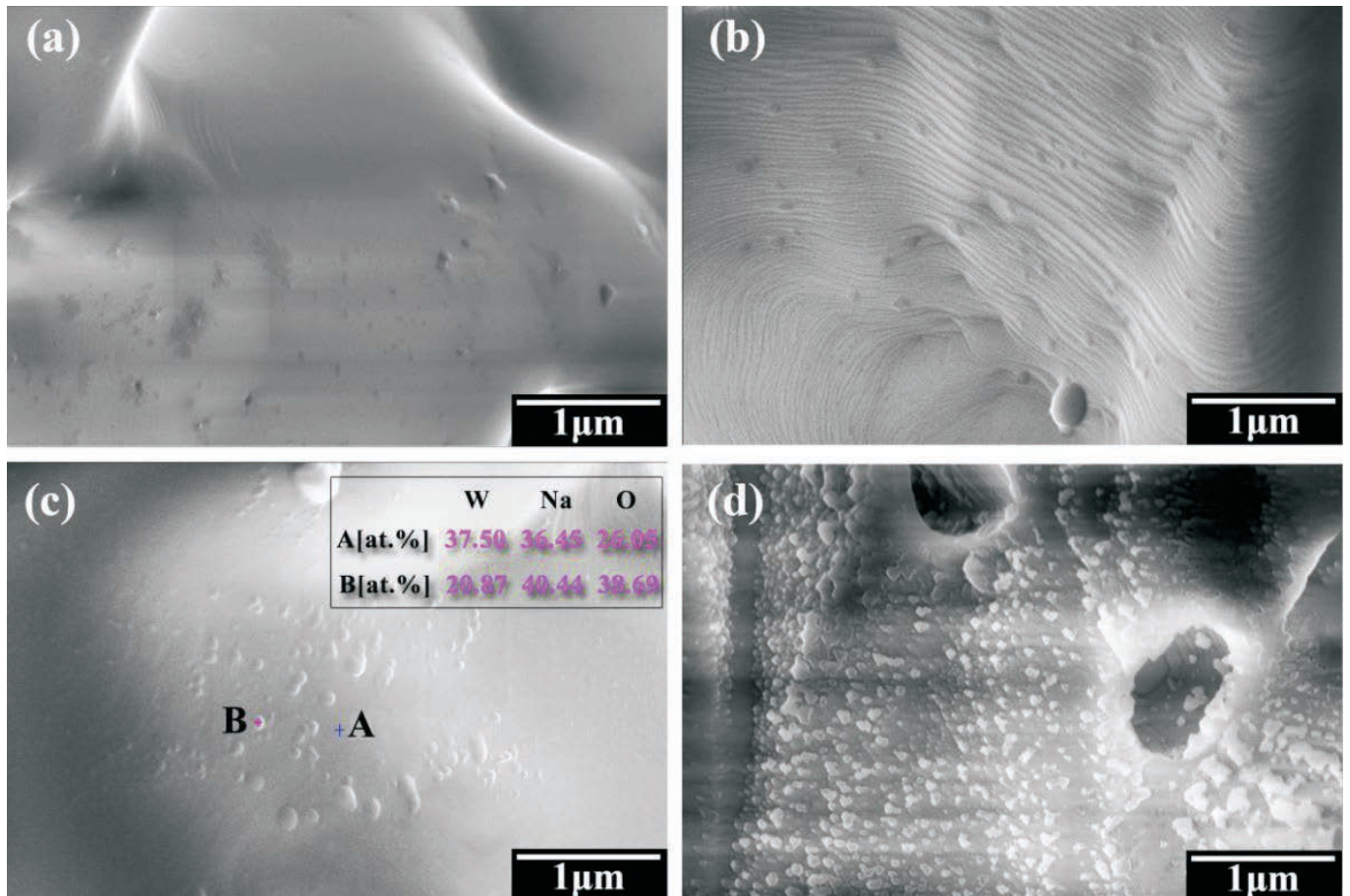


Fig. 2: SEM images of Na<sub>2</sub>WO<sub>4</sub> ceramic sintered at different temperature for 2 h, (a) 600 °C, (b) 650 °C, (c) 680 °C, (d) 700 °C.

increasing density, but in our study, the opposite was observed (Fig. 4a); this result is similar to that in Reference 15. On the basis of the XRD and SEM analysis, it could be determined that the formation temperature of the grains in  $\text{Na}_2\text{WO}_4$  ceramics was higher than  $650^\circ\text{C}$ . Consequently, crystallization plays a critical role in the variation of the dielectric constant.

It has been reported that  $Q \times f$  values at microwave frequencies depend on intrinsic and extrinsic factors, such as densification, impurities, substitution, grain boundaries, grain morphology and shape, secondary phase, pores, and so on<sup>16</sup>. Fig. 4b exhibits the change of the  $Q \times f$  value at different sintering temperatures, with this increasing clearly from 13 637 GHz at  $600^\circ\text{C}$  to 38 244 GHz at  $680^\circ\text{C}$ , indicating that increasing sintering temperature can assist in enhancing the  $Q \times f$  value of  $\text{Na}_2\text{WO}_4$  ceramics. However, the  $Q \times f$  value decreases with further increase in the sintering temperature, falling to 27 049 GHz at  $700^\circ\text{C}$ .

The  $\tau_f$  values of  $\text{Na}_2\text{WO}_4$  samples sintered at varying temperature from  $600$  to  $700^\circ\text{C}$  are shown in Fig. 4c. It can be seen that the  $\tau_f$  value decreased gradually from  $-36.5$  to  $-49.0$  ppm/K with increasing sintering temperature, indicating that the  $\tau_f$  values of  $\text{Na}_2\text{WO}_4$  ceramics should be adjusted to near zero with a composite or by adopting positive  $\tau_f$  for practical application in the future.

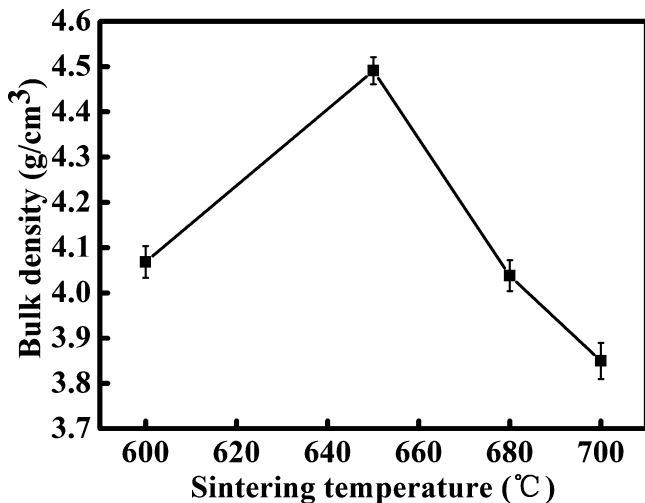


Fig. 3: Bulk density of the specimens sintered at varying temperature for 2 h.

The literature work and the current work were compared as shown in Table 1. It can be seen that  $\text{Na}_2\text{WO}_4$  ceramic is advantageous compared to other materials with regard to its low dielectric relative permittivity  $\epsilon_r$ , low dielectric loss and high  $Q \times f$  values after ultra-low temperature synthesis, which is a necessary requirement for microwave dielectric substrate applications. However, the  $\epsilon_r$  and  $Q \times f$  values of  $\text{Na}_2\text{WO}_4$  ceramics can be improved based on this study with the modified technology process, for example, with the conventional solid-state reaction method. Meanwhile its large negative  $\tau_f$  values should be improved to near zero before application via a method of solid solution and composites.

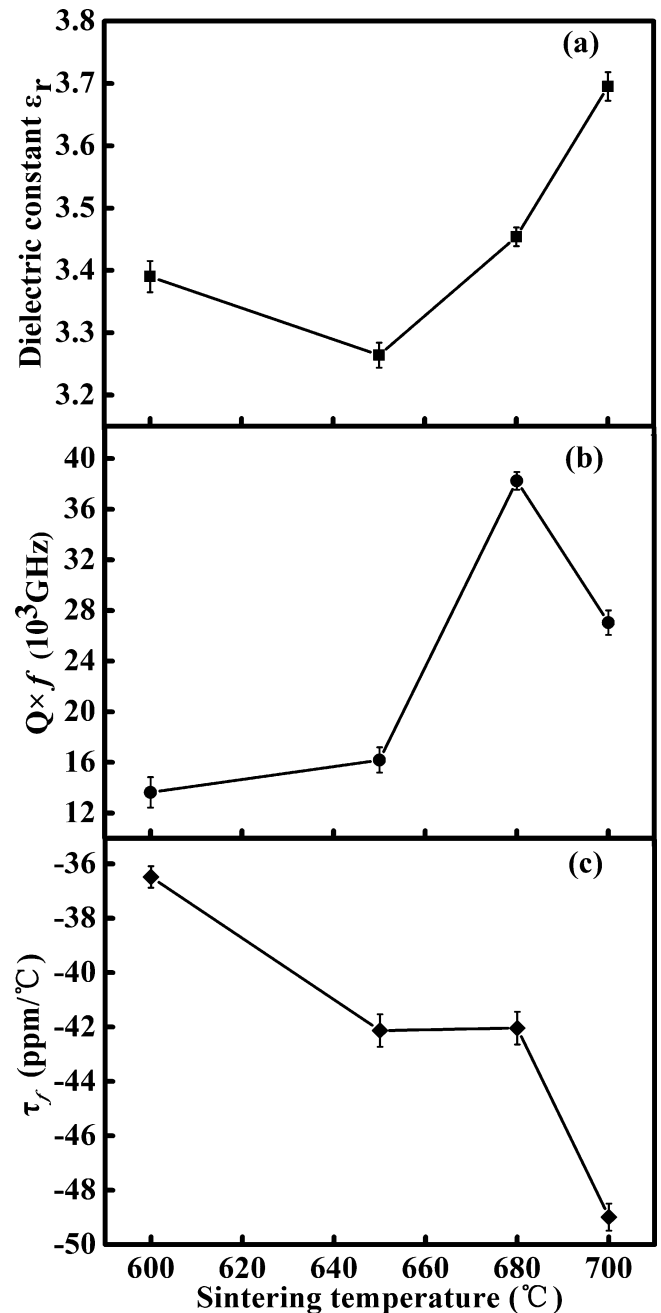


Fig. 4: The  $\epsilon_r$ ,  $Q \times f$ , and  $\tau_f$  values of  $\text{Na}_2\text{WO}_4$  ceramic sintered at varying temperature for 2 h.

#### IV. Conclusions

The cubic structure  $\text{Na}_2\text{WO}_4$  ceramic was prepared with the reactive sintering method using  $\text{Li}_2\text{CO}_3$  and  $\text{WO}_3$  powder as raw materials. The excellent microwave dielectric properties with a relative permittivity  $\epsilon_r$  of 3.45, a quality factor  $Q \times f$  of 38 244 GHz (at 13.64 GHz) and a temperature coefficient of resonant frequency  $\tau_f$  of  $-42.0$  ppm/K were obtained after sintering at  $680^\circ\text{C}$  for 2 h. The analytical results show that  $\text{Na}_2\text{WO}_4$  ceramic is a promising candidate for low-temperature ceramic technology.

#### Conflict of Interests

The authors declare that there is no conflict of interests regarding the publication of this paper.

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